Ion conductive characteristics of boron stabilized carbanion derived from organoboron n-conjugated systems

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Received: 20 January 2003/Revised version: 10 April 2003/ Accepted: 19 April 2003

Summary

Ion conductive properties of boron stabilized carbanions (BSC) have been studied as a model of π -conjugated ion conductive polymer in which counter anion of lithium ion is delocalized in π -conjugated system. After treatment of the BSC precursor (trimesitylborane or dimesityl-9-anthrylborane) with an organolithium reagent, the color of the reaction mixture dramatically turned to dark red or purple. Interestingly, BSC derived from dimesityl-9-anthrylborane showed moderate ion conductivities of 4.02×10^{-7} S/cm at 50° C even in bulk, indicating high dissociation degree of the BSC lithium salt. This can be attributed to the decreace in site hopping energy of the lithium ion, since anionic charge is delocalized via vacant p-orbital of the boron atom. These observations might have important implication for potential capability of ion conductive π -conjugated systems.

Introduction

Ion conductive materials have been intensely studied during the past few dacades in view of application for lithium secondary batteries as electrode separater. In particular, poly(ethylene oxide) derivatives [1] have been attractive research target due to their excellent ability to dissolve a variety of salts and transport the generated carrier ions. However, such ionic conduction dependent on the segmental motion of the polymer chain should, in principle, not exceed the conductivity of the corresponding model viscous solution $({\sim}10^{4}S/cm)$.

Therefore, in order to create ionic conductors with conductivity several orders of magnitude higher than are studied today, novel ion conductive materials with a different conduction mechanism must be sought after. Moreover, such materials whose conduction is independent of segmental motion of electrolyte matrix are expected to show much higher selectivity for lithium ion transport.

As a novel approach for the preparation of ion conductive materials, one of the authors (H. Ohno) advocated the utilization of an n-type conjugated polymers as lithium ion conductive material [2]. In such materials, the lithium counter ion is delocalized in a π -conjugated system to reduce site hopping energy of lithium ion (Figurel). If only a long range electric conduction was prohibited by control of interchain distance (via polymer assembly or control of orientation, etc.), such materials would have a considerable potential for fast and selective lithium ion transport.

Fig.1 Schematic illustration for stabilization of cation with π -conjugated system

As a candidate for such a material, lithium intercalated polyacetylene [3] has been prepared by **an** electrochemical method, and has been studied extensively as an electrode material. Although metallic conductivity is reported for this material, the contribution of lithium ion conduction has not yet been clarified. Futhermore, such electrochemical doping of π -conjugated system includes generation of a variety of radical carrier species. This is unfavorable, since most of π -conjugated systems are not stable in the n-doped state.

For the preparation of selectively anion-dopable π -conjugated systems, Leclerc et al. recently reported the synthesis and conductive properties of aniondopable ester modified polyfluorene derivatives (41. Their results showed that aniondoped π -conjugated systems without radical carrier species also exhibited high conductivity. However, electronic conduction was not prohibited in their system either and the contribution of lithium ion conduction remained unclear.

In this study, we have investigated ion conductive characteristics of some model compounds to assess the effect of π -conjugation on the ionic conductivity without contribution of electronic conduction. As a model for this study, some boron-stabilized-carbanions (BSC) *[5]* were prepared and their ion conductive characteristics were studied. Recently, organoboron conjugated systems [6, 71 are attracting growing interest due to their unique electronic structure with extended π conjugation through the vacant p-orbital of the boron atom. The pioneering work by Y. Chujo et al [6]. demonstrated that π -conjugated organoboron polymers prepared by hydroboration polymerization [6a] possess high electron affinity, from cyclic voltammetry and semi-empirical calculation besed on Hartly-Fock theory IS]. From these results, the organoboron π -electron system was considered to be a good candidate as a novel type of anion-dopable π -electron system with a stable n-doped state and a high degree of dissociation of the lithium ion.

Experiin ental Section

Poly(ethyleneglycol) dimethylether were dried over molecular shieves before use. Tetrahydrofuran was dried over sodium and distilled before use. Trimesitylboron, bromoanthracene was purchased from Aidrich and used without further purification. nBuLi in hexane (1.60N) was purchased from Kanto Chemicals Co. Ltd. Dimesitylarylboranes were prepared according to reported procedure [S]. The BSC formation of aryldimesitylborane was examined in a simmilar manner with the case of Mes,B, reported by B. G. Ramsey et. a1 *[5].* After the removal of solvent, resulting purple solid was dried in *vacuo* before subjected to measurement of conductivity. The measurements of ¹H- and ¹¹B-NMR spectra were carried out on The measurements of ¹H- and ¹¹B-NMR spectra were carried out on $JEOL-\alpha500$.

The ionic conductivities for organoboron polymer electrolytes were evaluated by the complex impedance method on a Solartron inadel 1260 (Schlumberger) impedance analyzer with an ac frequency range of 1Hz to 1MHz.

Results and Discussions

The preparation of BSC derived from trimesitylborane *[5]* was reported by B. G. Ramsey et al. and charge stabilization effect in the BSC was discussed. Although BSC was found to contain intimate ion pairs in THF or DMSO, substantial charge delocalization to boron was implied by a ${}^{11}B\text{-NMR}$ study, which would reveal a close coordination of the cation to a specific site.

The BSC1(derived from trimesitylborane) was prepared according to the reported procedure, by treating trimesitylboron with n-BuLi in THF solution for several hours (Scheme la). After removing the solvent and drying in vacuo, the resulting purple solid was subjected to measurement of ionic conductivity by the ac impedance method.

The temperature dependence of the conductivity for BSCl is depicted in Figure 2. BSC1 showed a conductivity of 8.17×10^{-8} S/cm at 50°C even in the bulk. Although the coductivity was low, it was demonstrated that site hopping of the lithium ion between n-electron systems was possible in spite of the prohibited electronic conduction in this model system.

Scheme I a) Preparalion of BSC precursors b) BSC formation

Fig.2 Temperature dependence of ionic conductivity for BSCl and BSCZ in the presence and absence of PEO, respectively.

lOOO/T(1/K)

To assess the effect of π -conjugation degree on the ionic conductivity, the preparation of some BSCs having a naphtyl or anthryl group was attempted. According to Scheme 1, each BSC precursor was prepared by reaction of aryl lithium with dimesitylfluoroborane in 61~66% yields. Then dimesityl-1-naphtylborane was treated with nBuLi under the same condition with the case of trimesitylboron. However, the characteristic color change was not observed, indicating that no BSC was formed. In the ¹¹B-NMR spectrum of the resulting material, the main peak was observed at -7.71 ppm. This suggests that borate was formed possibly because of a smaller steric hindrance around the boron atom. On the other hand, a typical color change to deep red was observed when BSC2 was formed from dimesityl-9 anthtylborane. The "B-NMR spectrum showed an additional peak at 38.3 ppm (Figure 3) corresponding to BSC. From the integration ratio of the peak to that for BSC precursor, it was estimated that 20.5% of the starting material was converted to BSC2. After removal of solvent and drying in vacuo, the ionic conductivity of thc soft solid was measured. The conductivity at 50° C was 4.02×10^{-7} S/cm, one-order of magnitude higher than that for BSC1. Since electronic conduction is prohibited in these model systems, the increace in conductivity is ascribed to a markedly reduced site hopping energy of the lithium ion in these systems. This might be due to a further delocalization of anionic charges in the extended π -conjugated system via empty p-orbital of the boron atoms.

The conductivities of BSC 1 and 2 in the presence of PEO were also evaluated as shown in Figure *2.* Since BSCs are unstable in the presence of protic solvent, PEO dimethyl ether (Mw 400) was used as polymer solvent to dissolve BSCs.

The conductivities increased in PEO possibly due to assistance of segmental motion of the matrix, and further dissociation of BSC salt in the polar environment. However, observed conductivity was not particularly high as PEO-lithium salt complex, due to limited solubility of BSCs in PEO.

Fig.3 "B-NMR spectra of dimesitylanthrylborane and BSC2

In summary, ion conductive properties of BSCs were investigated by the ac impedance method. Even though the conductivity was not very high, significant conductivity was observed even in bulk. When introducing the anthryl group, the conductivity of BSC was improved by one-order of magnitude. This is possibly owing to a decrease of the site hopping energy of the lithium ion in the larger π conjugated system. The bulk conductivities of BSCs are not very high at present, however, it should be noted that bulk conductivity was observed even the case of model compounds with very limited conjugation length despite the prohibited electric conduction. This phenomenon may have important implications for the creation of a novel type of ion conductive polymers.

Acknowledgements. The present study was carried out under the COE program of Future Nano-Materials. This work is financially supported by The MURATA SCIENCE FOUNDATION., and a Grant-in-Aid for Scientific Research from the Ministry of Education, Science, Sports, and Culture, Japan (No. 14205136)

References

- 1. (a) Fenton DE, Parker JM, Wright PV (1973) Polymer 14:589 (b) Wright PV, (1975) Br Polym J 7:319 (C) Wright PV (1976) J Polym Sci Polym Phys Ed 14:955
- 2. Ohno H (1990) Kobunshi 39(5):357
- 3. (a) Chiang CK (1981) Polymer 22:1454 (b) Fouletier M, Degott P, Armand MB (1983) Solid State Ionics 8:165
- 4. (a) Ranger M, Leclerc M (1997) J Chem Soc Chem Commun 1997:1597 (b) Ranger M, Rondeau D, Leclerc M (1997) Macromolecules 30:7686 (c) Ranger M, Leclerc M (1999) Macromolecules 32:3306
- *5.* Ramsey BG, Isabelle LM (1981) J Org Chem46(1):179
- 6. (a) Matsumi N, Naka K, Chujo Y (1998) J Am Chem SOC 120:5112 (b) Matsumi N, Naka K, Chujo Y (1998) J Am Chem Soc 120:10776 (c) Matsumi N, Miyata M, Chujo Y (1999) Macromolecules 32:4467 (d) Matsumi N, Umeyama T, Chiijo Y (2000) Macromolecules 33:3956 (e) Matsumi N, Chujo Y (2000) Contemporary Boron Chemistry (Ed: Davidson MG, Hughes AK, Marder TB, Wade K; Royal Sociely of Chemistry, UK) 51.
- 7. (a) Corriu R J-P, Deforth T, Douglas WE, Guerrero G, Siebert WS (1998) Chem Commun 963 (b) Noda T, Shirota Y (1998) J Am Chem Soc 120:9714 (c) Noda T, Ogawa H, Shirota Y (1999) Adv Mater 11:283 (d) Yamaguchi S, Akiyama *S,* Tamao K (2000) J Am Chem Soc 122:6335 (e) Yamaguchi S, Akiyama S, Tamao K (2001) J Am Chem Soc 123: 11 372
- 8. (a) Chujo Y, Matsumi N (2000) Polym Prep(ACS Division of Polymer Chemistry) 41(1):549 (b) Matsumi N, Chujo Y , Okada K, Tanaka K unpublished data.